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SEMICONDUCTOR ELECTRODES 25. THE P-GAAS/HEPTYL VIologen SYSTEM.

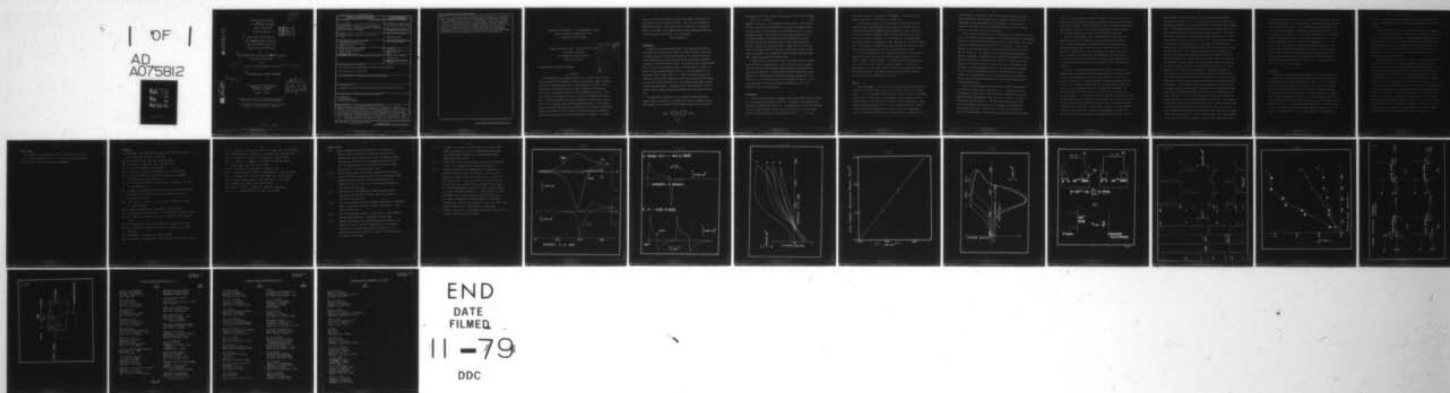
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Task No. NR 051-693

TECHNICAL REPORT No. 2

LEVEL

6 Semiconductor Electrodes, 25.  
The p-GaAs/Heptyl Viologen System.  
Photoelectrochemical Cells and  
Photoelectrochromic Displays.

by

10 Benjamin Reichman, Fu-Ren F. Fan and Allen J. Bard

Prepared for Publication

in the

12 30 Journal of the Electrochemical Society

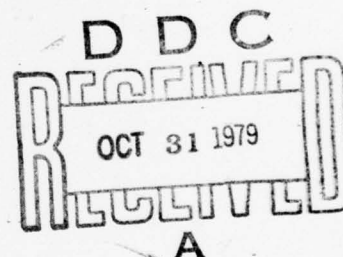
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9 Technical rept. 1 Sep 79-31 Aug 80,

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The University of Texas at Austin  
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REPORT DOCUMENTATION PAGE		READ INSTRUCTIONS BEFORE COMPLETING FORM
1. REPORT NUMBER 2	2. GOVT ACCESSION NO.	3. RECIPIENT'S CATALOG NUMBER
4. TITLE (and Subtitle) Semiconductor Electrodes 25. The p-GaAs/Heptyl Viologen System. Photoelectrochemical Cells and Photoelectrochromic Displays.		5. TYPE OF REPORT & PERIOD COVERED 1 Sept. 1979 to 31 Aug. 1980
7. AUTHOR(s) Benjamin Reichman, Fu-Ren F. Fan and Allen J. Bard		6. PERFORMING ORG. REPORT NUMBER
9. PERFORMING ORGANIZATION NAME AND ADDRESS Department of Chemistry University of Texas at Austin Austin, TX 78712		8. CONTRACT OR GRANT NUMBER(s) N00014-78-C-0592
11. CONTROLLING OFFICE NAME AND ADDRESS Office of Naval Research 800 N. Quincy Arlington, VA 22217		10. PROGRAM ELEMENT, PROJECT, TASK AREA & WORK UNIT NUMBERS
14. MONITORING AGENCY NAME & ADDRESS (if different from Controlling Office)		12. REPORT DATE October 1, 1979
		13. NUMBER OF PAGES 30
		15. SECURITY CLASS. (of this report) Unclassified
		15a. DECLASSIFICATION/DOWNGRADING SCHEDULE
16. DISTRIBUTION STATEMENT (of this Report) This document has been approved for public release and sale; its distribution is unlimited.		
17. DISTRIBUTION STATEMENT (of the abstract entered in Block 20, if different from Report)		
18. SUPPLEMENTARY NOTES Prepared for publication in the Journal of the Electrochemical Society.		
19. KEY WORDS (Continue on reverse side if necessary and identify by block number) semiconductors photoelectrochemistry electrochromic displays		
20. ABSTRACT (Continue on reverse side if necessary and identify by block number) The photoreduction of aqueous solutions of 1,1'-diheptyl-4,4'-bipyridyl (heptyl viologen, $HV^{2+}$ ) bromide on p-GaAs was investigated. Under irradiation with light of wavelengths below 885 nm, the photoreduction of $HV^{2+}$ to form a precipitate of violet $HVBr$ occurs on p-GaAs at a potential of $\sim 400$ mV more positive than the reversible potential of the $HV^{2+}/HVBr$ couple on Pt. Oxidation of $HVBr$ back to soluble $HV^{2+}$ occurs in the dark at more negative potentials. Photoelectrochemical low capacity storage cells based on this reaction		

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S/N 0102-014-6601Unclassified  
SECURITY CLASSIFICATION OF THIS PAGE (When Data Entered)

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Semiconductor Electrodes 25. The p-GaAs/Heptyl Viologen  
System. Photoelectrochemical Cells and  
Photoelectrochromic Displays.

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(ABSTRACT)

The photoreduction of aqueous solutions of 1,1'-diheptyl-4,4'-bipyridyl (heptyl viologen,  $HV^{2+}$ ) bromide on p-GaAs was investigated. Under irradiation with light of wavelengths below 885 nm, the photoreduction of  $HV^{2+}$  to form a precipitate of violet HVBr occurs on p-GaAs at a potential of  $\sim 400$  mV more positive than the reversible potential of the  $HV^{2+}/HVBr$  couple on Pt. Oxidation of HVBr back to soluble  $HV^{2+}$  occurs in the dark at more negative potentials. Photoelectrochemical low capacity storage cells based on this reaction with Cu, Ni, or  $H_xNb_2O_5$  counter electrodes were constructed. In such cells, under illumination, the p-GaAs electrode behaves as the cathode, a photovoltage develops and current flows in the external circuit, while HVBr precipitates on the electrode surface and the counter electrode is oxidized. In the dark,

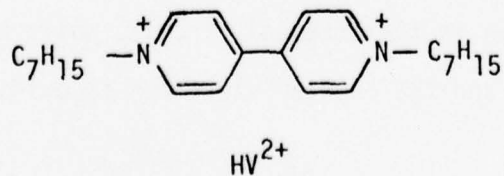
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(End of Abstract)

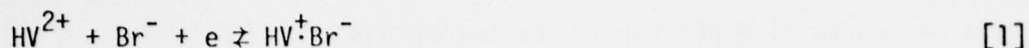
### INTRODUCTION

A number of photoelectrochemical (PEC) cells employing semiconductor electrodes for the conversion of radiant to electrical energy (liquid junction photovoltaic cells) have been described (1-17). Most have employed n-type semiconductors and have been used for the instantaneous production of electricity without chemical storage. The possibility of a photochargeable PEC cell using a third storage electrode or with production of soluble reactants has been discussed however (8,10-11). PEC cells employing p-type electrodes are less common (8-9,15-17). Studies of aqueous systems with p-GaAs and p-GaP have been reported, (15-16,18) but the observed efficiencies of these were usually quite small and in many cases problems with electrode stability were encountered. Investigations of p-GaAs electrodes in acetonitrile solutions suggest good stability under strict water-free conditions (19).

We report here investigations of the p-GaAs electrode immersed in aqueous solutions containing 1,1'-diheptyl-4,4'-bipyridyl (heptyl viologen,  $HV^{2+}$ ) bromide. The reduction of  $HV^{2+}$  bromide on metal electrodes yields a



precipitate of the violet radical cation salt (eq. [1]) and this has been



suggested as the basis of electrochromic displays (20-22). The formation of such a precipitate by photoreduction at a p-type semiconductor would allow low capacity energy storage via a solid material, just as occurs in secondary batteries. We will show that such a photoreduction does occur at p-GaAs at potentials  $\sim 400$  mV less negative than that observed at a platinum electrode and that dark oxidation of the HVBr precipitate is possible. Moreover the p-GaAs electrode is stable both in the dark and under illumination in the  $\text{HV}^{2+}$  solution so that PEC cells using suitable counter electrodes (e.g., Cu/CuO) which show a periodic electrical output under chopped illumination can be constructed.

A second aspect of these studies concerns the application of reaction [1] on p-GaAs for display purposes. Several devices for active displays based on combining electrogenerated chemiluminescence and semiconductors have been described (23). A passive photoactivated display based on metal deposition on a semiconductor electrode was recently reported (24). The principles of a similar display based on precipitation of HVBr at a point on a semiconductor electrode simultaneously addressed by light and an electrical signal is described here.

#### EXPERIMENTAL

The single crystal p-GaAs was obtained from Atomergic Chemicals (Long Island, N.Y.) with acceptor concentration  $3 \times 10^{18} \text{ cm}^{-3}$  (Zn-doped) and was mounted as an electrode as previously described (13). The surface (face 111) was first polished with  $0.5 \mu\text{m}$  alumina powder and then etched for 10-15 sec in a solution which contained  $\text{H}_2\text{SO}_4:\text{H}_2\text{O}:30\% \text{H}_2\text{O}_2$  in concentration ratio 3:1:1. In some

experiments single crystal p-GaP was also employed. The surface of this crystal was etched in 11 M HCl for 10 sec before use.

As light sources either a 450 W xenon lamp or a He-Ne laser (1.6 mW power) was used. The potentiostatic measurements were carried out with a Princeton Applied Research (PAR) potentiostat Model 173, with a cell consisting of an aqueous saturated calomel reference electrode (S.C.E.) and a Pt wire as the counter electrode. The electrochemical cell was provided with pyrex windows for the passage of light. Nitrogen gas was passed through the solution before the experiments and above the solution during the experiments, since viologen radical cations are very sensitive to oxygen.

In the two electrode PEC cell experiments the voltage of the photocell between the p-GaAs and the oxide counter electrode was measured on a x-t recorder through a voltage follower. The photocell current was measured on an x-t recorder through a current-to-voltage-converter. For short time measurements of the cell short circuit current, a Nicolet Model 1090A digital oscilloscope was used. The light intensity at the semiconductor surface was varied by means of neutral density filters.

## RESULTS

Cyclic voltammetry. The i-V curve at a p-GaAs electrode of an unstirred solution of 0.01 M  $HV^{2+}$  and 0.2 M KBr at a scan rate of 100 mV/sec is shown in Fig. 1. In the dark (curve a) there was only a small current in the cathodic region with a very small peak on scan reversal. Under illumination of the p-GaAs electrode surface with the full power of the 450 W xenon lamp, a larger cathodic current beginning at  $\sim 0.1$  V vs. S.C.E. appeared (curve b). Simultaneously with the appearance of this cathodic current, the precipitation of the violet  $HVBr$  was observed on the electrode surface. On reversing the



scan direction, a sharp anodic current peak occurred, either in the dark or under illumination. During the passage of the anodic current the violet precipitate on the semiconductor surface disappeared. Thus the overall process is that shown in [1], with illumination necessary for the reduction process, as is usual for p-type semiconductors.

These cyclic voltammograms could be repeated many (at least 100) times with only a small change in the peak height of the cathodic photocurrents, demonstrating good chemical reversibility of the deposition and stripping process. Upon illuminating the electrode with the narrow beam of a He-Ne laser, violet spots appeared on the electrode surface at exactly the location and of the size of the laser beam spot. The response time of the process was investigated by potential steps under illumination or by irradiation with chopped light with the electrode held at a constant potential (Fig. 2). A good coloration was observed on the electrode surface within  $\sim 0.5$  sec after illumination of the electrode. Although the intensity ratio for HVBr deposited on dark p-GaAs is not very high, the deposition was easily discernable and this type of process can be considered for use as an electrochromic display controlled by both light and electrode potential and may also be applicable to photoimaging purposes (24-25).

The cyclic voltammograms of the solution of  $0.01 \text{ M HV}^{2+}$  dibromide and  $0.2 \text{ M KBr}$  at a Pt electrode is shown in Fig. 1c. By comparison of this voltammogram with that obtained with p-GaAs under illumination, one can see that significant photoreduction of  $\text{HV}^{2+}$  can occur on p-GaAs at potentials  $\sim 400 \text{ mV}$  less negative than on Pt. The photocurrent attributable to the reduction of protons on the p-GaAs in a KBr solution not containing  $\text{HV}^{2+}$  was very small (Fig. 1d). Photoreduction of  $\text{HV}^{2+}$  was also investigated with a p-GaP photocathode. In this case the underpotential for the reduction



process was even larger than in the case of p-GaAs, but the process was not very reversible and the back oxidation occurred only at more positive potentials than on a Pt electrode. Thus upon scanning a p-GaP electrode over roughly the same potential region (Fig. 3), a gradual accumulation of HVBr occurred on the electrode surface and the cathodic current gradually decreased.

With the light intensity used in the experiments described above the photoreduction current on p-GaAs was limited by the rate of diffusion of  $HV^{2+}$  to the electrode surface, as demonstrated by the proportionality of the cathodic photocurrent to the square root of the scan rate,  $v$  (Fig. 4). Moreover, as shown in Fig. 5, while the potential for photocurrent onset and the current at the foot of the wave depended upon the light intensity, at more negative potentials the maximum current was almost independent of intensity and was limited by the rate of diffusion of  $HV^{2+}$  to the electrode surface.

PEC cells. Since the photoreduction of  $HV^{2+}$  occurs on p-GaAs with an underpotential of about 400 mV while the back-oxidation in the dark on p-GaAs occurs at the same potentials as on Pt, one can envision a sort of solar storage cell composed of a p-GaAs and another chemically reversible electrode (Fig. 6). This counter electrode should have a redox potential between the potential for the photoreduction of  $HV^{2+}$  on p-GaAs and the potential for oxidation of HVBr in the dark (Fig. 6c). In such a cell, when light irradiates the p-GaAs, it is the cathode of the cell and HVBr precipitates (Fig. 6a). The HVBr precipitate on the p-GaAs is stable at open circuit and can be oxidized in the dark back to the original  $HV^{2+}$  while the counter electrode becomes the cathode of the cell (Fig. 6b). Three counter electrodes with redox potentials suitable for such a photocell, Cu/Cu-oxide,  $H_xNb_2O_5/Nb_2O_5$ , and Ni/Ni-oxide, were used. The open circuit voltages and the short circuit currents of such

cells composed of p-GaAs and the three counter electrodes under periodic illumination are shown in Fig. 7. The relative polarity of the p-GaAs shows the expected changes (Fig. 6) as the semiconductor is subjected to the chopped beam. In the photocharged condition, the HVBr remained on the electrode surface and the open circuit voltage was stable for at least 2 hr. The maximum open circuit voltages were found with the Cu counter electrode with potential excursions of up to 0.4 V observed (Fig. 7A). The short circuit currents of these cells are shown in Fig. 7B. In the light, current flows in the external circuit during which precipitation of the purple HVBr on the surface of the p-GaAs occurs. The current decays almost to zero in about 10 sec because a thick opaque film of HVBr builds up on the electrode surface and the  $HV^{2+}$  concentration is depleted near the electrode surface. The cell can be regarded as a storage battery. The precipitate of HVBr can be stored on the p-GaAs and then oxidized in the dark to yield a current in a direction opposite to that observed under illumination. Obviously the capacity of such a cell is very small; the maximum charge in the form of HVBr which can precipitate on the p-GaAs before the current drops to zero is about  $10 \text{ mC/cm}^2$ . Alternately, the cell can be considered as a type of a.c. device under periodic illumination. In this mode the p-GaAs cell performance was very stable with all three cells and no deterioration or decrease in the current were observed after 4 days of continuous operation with 20 sec period illumination by the 450 W xenon lamp. The open circuit voltage, the current and the amount of charge involved were all reproducible and stable to within 10%. The variation of the initial short circuit current on light intensity under irradiation with the He-Ne Laser (1.6 mW power) was examined with all three cells (Fig. 8). A linear dependence of the short circuit current on light intensity was observed at the lower intensity levels. In the case of Cu counter electrode, saturation of the current occurred at  $\sim 2.5 \text{ mA/cm}^2$ . With Ni as the counter electrode, no

saturation of the short circuit current occurred even at the highest intensity used, while with  $\text{Nb}_2\text{O}_5$  as a counter electrode a linear dependence was observed only at very low light levels. In all cases the currents obtained were the highest for the Cu electrode. The current saturation observed in this case can probably be ascribed to the limitation by diffusion of  $\text{HV}^{2+}$  to the p-GaAs surface, since the magnitude of the saturation current density in Fig. 8a is similar to the limiting current found in the voltammograms, Fig. 1b, with identical  $\text{HV}^{2+}$  concentrations. With the  $\text{Nb}_2\text{O}_5$  electrode the short circuit current reaches saturation earlier than in the case of Cu and was clearly limited by the oxidation reaction taking place on the  $\text{Nb}_2\text{O}_5$  electrode. The maximum quantum efficiencies for the photocells were calculated from the slopes of the linear portions of the curves in Fig. 8, yielding values of  $\sim 20\%$  (Cu),  $\sim 10\%$  (Ni), and  $2\%$  ( $\text{Nb}_2\text{O}_5$ ).

#### DISCUSSION

The behavior of the p-GaAs/solution interface and a rationalization of the observed PEC cell behavior can be presented in terms of the surface controlled model for GaAs proposed elsewhere (26) (Fig. 9). In this model the Fermi level of the semiconductor is pinned by surface states located about one-third of the way up in the gap (i.e.,  $\sim 0.4$  to  $0.5$  V above the valence band edge) (Fig. 9a). The maximum open circuit photopotential,  $V_{\text{on}}$ , observed under intense illumination would then be  $\sim 0.4$  V, and the open circuit photovoltage would be the difference between  $V_{\text{on}}$  and  $V_{\text{redox}}$  of the counter electrode reaction (Fig. 9b). In the dark after deposition of  $\text{HVBr}$ , oxidation of  $\text{HVBr}$  occurs via the surface states and the p-GaAs essentially behaves as an inert contact to the layer (Fig. 9c). The counter electrode in a two-electrode

PEC cell must have a redox potential between  $V_{on}$  and that of the  $HV^{2+}/HVB$  system, so that only couples in a relatively narrow range of potentials are suitable.

Storage cells based on this type of electrode reaction will necessarily have small capacities unless films which deposit on the electrode surface can be found which are both transparent to the wavelengths absorbed by the semiconductor and are electrical conductors. Moreover higher concentrations of the soluble form are required to prevent mass transfer limitations to the photocurrent even before film formation causes current decay. Such cells may however find application in very low capacity batteries or as a.c. devices under periodic illumination.

The application of such a system to produce a display addressable both optically and electrically (a photoelectrochromic display) is perhaps of greater interest (Fig. 10). One can envision the operation of such a device in which a light beam produces a raster on the electrode surface and the electrode potential is switched to (a) the deposition potential (coloring), (b) open circuit (hold), or (c) the stripping potential (erase). The advantage of such a display compared to the more familiar electrochromic ones is that only two leads need be provided to the cell rather than the multiple leads and complicated electrical addressing required for a large screen or multi-character display. While the contrast ratio for  $HVB$  on  $GaAs$  is not very good, in principle a wide band gap semiconductor ( $E_g > 3$  eV), which would appear white and would be addressed with an ultraviolet beam could be employed. For a display involving rapid scanning, a system with a faster response time is also required. Similar concepts involving semiconductors and metal depositions for photoimaging have been reported (24, 27).

ACKNOWLEDGMENT

The support of this research by the Office of Naval Research, which is a joint project with Professor A.B.P. Lever of York University, and the National Science Foundation is gratefully acknowledged.



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# FIGURE CAPTIONS

- Fig. 1 Cyclic voltammograms recorded with p-GaAs at 100 mV/sec with aqueous solution containing 0.01 M  $\text{HVBr}_2$  and 0.2 M KBr (a) in the dark; (b) under illumination; (c) cyclic voltammogram recorded with the same solution on a Pt disk electrode (100 mV/sec); (d) cyclic voltammogram recorded with p-GaAs in solution containing only 0.2 M KBr. Light source, 450 W Xe lamp.
- Fig. 2 Current-time transients in potential step experiments on p-GaAs in solution containing 0.01 M  $\text{HVBr}_2$  and 0.2 M KBr (a) potential stepped between -0.5 and +0.1 V under constant illumination; (b) the potential held at -0.35 V vs. S.C.E. under chopped illumination. Light source, 450 W Xe lamp.
- Fig. 3 Consecutive cyclic voltammograms at irradiated p-GaP solution containing 0.01 M  $\text{HVBr}_2$  and 0.2 M KBr. Scan rate, 100 mV/sec. Light source, 450 W Xe lamp.
- Fig. 4 Scan rate  $v$  dependence of the peak cathodic photocurrent at irradiated p-GaAs in a solution containing 0.1 M  $\text{HVBr}_2$  and 0.2 M KBr. Light source, 450 W Xe lamp.
- Fig. 5 Cyclic voltammograms at p-GaAs in solution of Fig. 4 with different relative light intensities. Light source, 450 W Xe lamp.
- Fig. 6 Schematic description of the  $\text{HVBr}_2$ -based photoelectrochemical cell operating (a) under illumination (photocharge); (b) in the dark (discharge); (c) relative energy levels and potentials at the counter electrode and at the p-GaAs, required for operation of the cell shown in (a) and (b).

- Fig. 7 (A) Open circuit potentials and (B) short circuit current under chopped illumination for the cells (a) p-GaAs/HVBr/HVBr<sub>2</sub>.KBr/Cu; (b) p-GaAs/HVBr/HVBr<sub>2</sub>.KBr/Nb<sub>2</sub>O<sub>5</sub>; (c) p-GaAs/HVBr/HVBr<sub>2</sub>.KBr/Ni. Light source, 450 W Xe lamp.
- Fig. 8 Initial short circuit photocurrents for the three cells described in Fig. 7 as a function of light intensity (He-Ne laser).
- Fig. 9 Schematic description of the energy levels of the HVBr<sub>2</sub> based photoelectrochemical cells in the light at (a) short circuit, no HVBr film on p-GaAs initially and (b) open circuit, no HVBr film on p-GaAs initially. In the dark at (c) short circuit, with HVBr film on p-GaAs and (d) open circuit, with HVBr film on p-GaAs.  $E_c$  = conduction band edge of p-GaAs;  $E_f$  = Fermic energy of p-GaAs;  $E_v$  = valence band edge of p-GaAs; s.s. = surface state;  $E_v^0$  = energy corresponding to the redox potential of heptyl viologen couple;  $E_c^0$  = energy corresponding to the redox potential of the counter electrode;  $V_{ph}$  = open circuit voltage in the light;  $V_{dk}$  = open circuit voltage in the dark;  $V_{on}$  = onset potential of photocurrent.
- Fig. 10 Schematic description of a photoelectrochromic display device based on HVBr<sub>2</sub> reduction on p-GaAs electrode.

Fig. 1

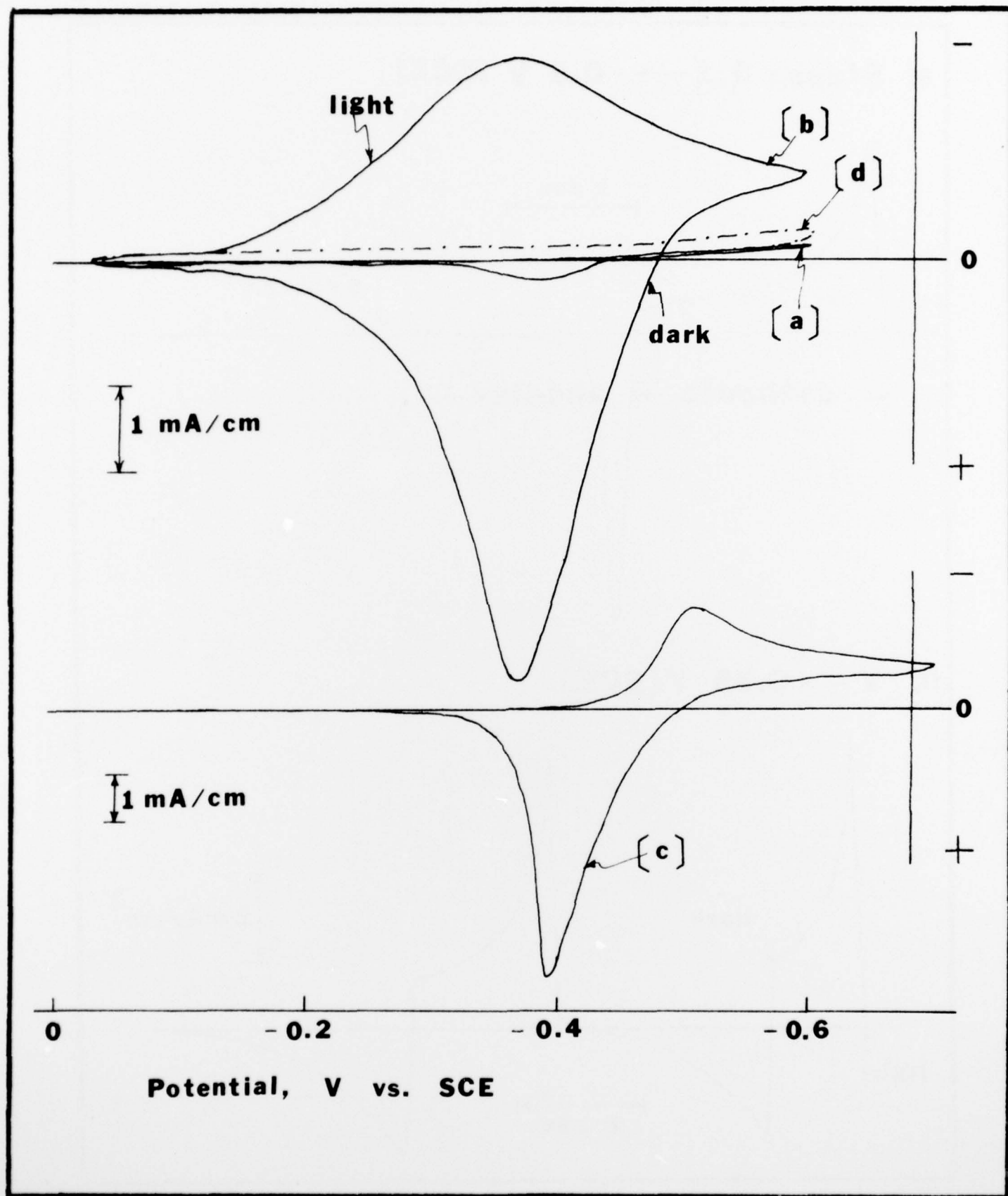
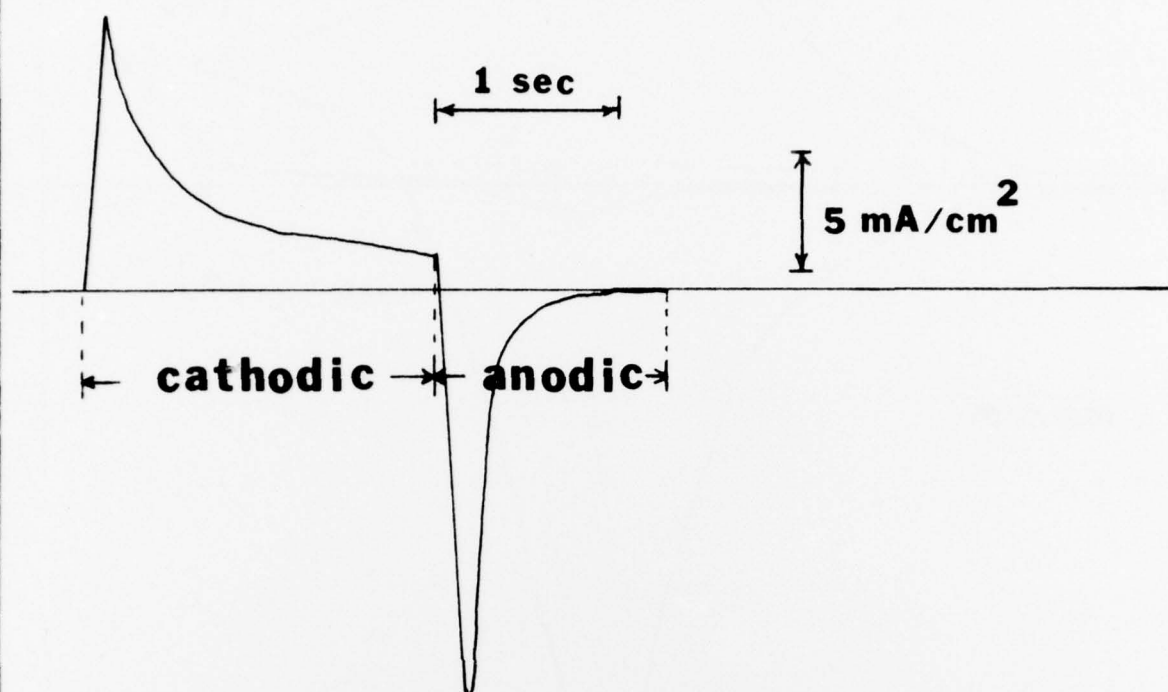




Fig 2

a. Steps:  $0.1 \leftrightarrow -0.5$  V (SCE)



b.  $V = -0.35$  V (SCE)

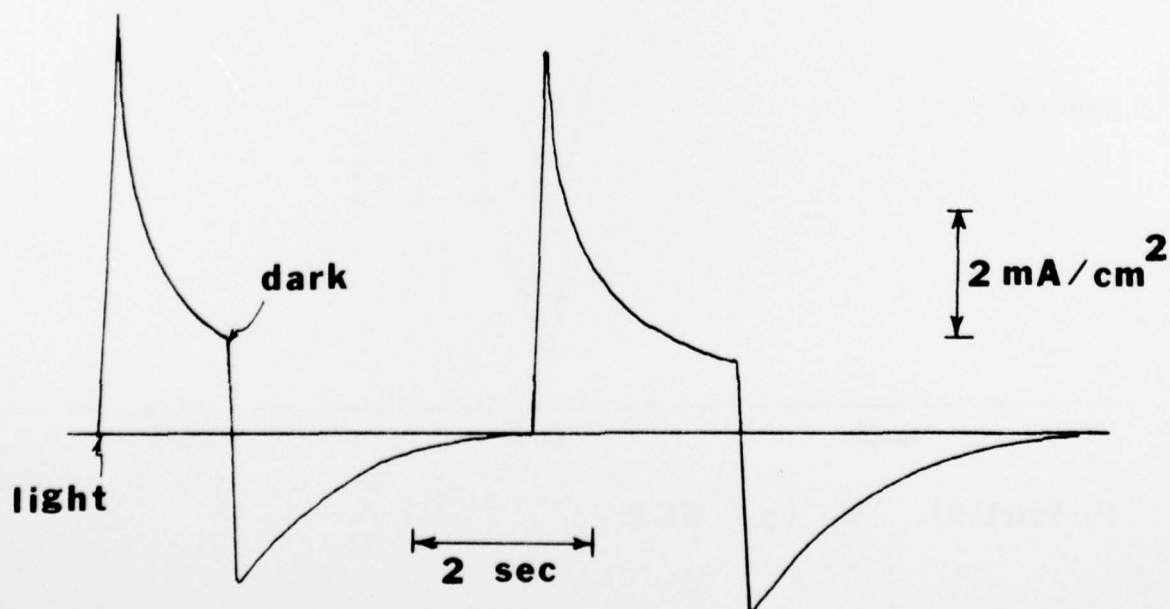


Fig. 3

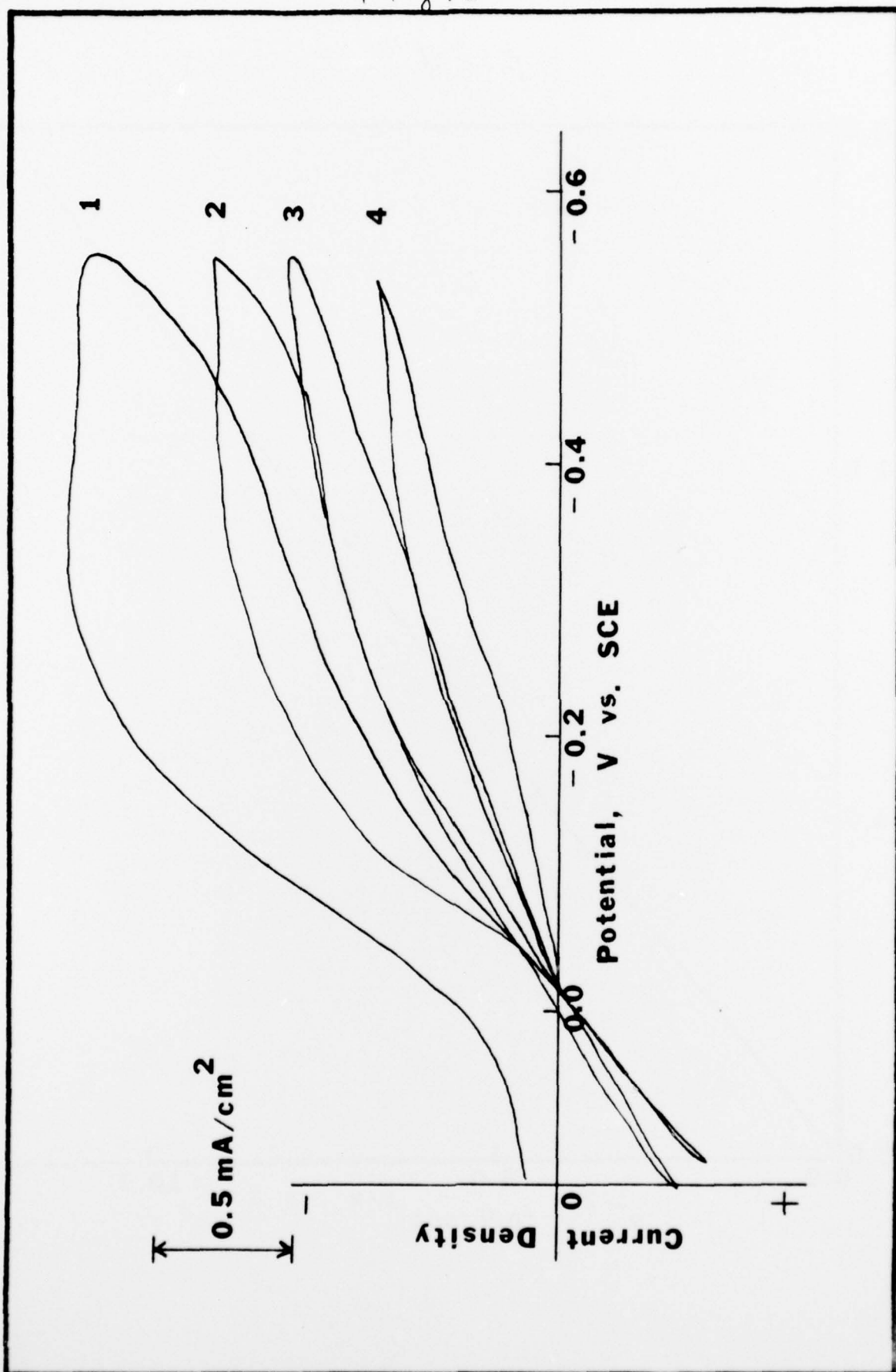


Fig. 4

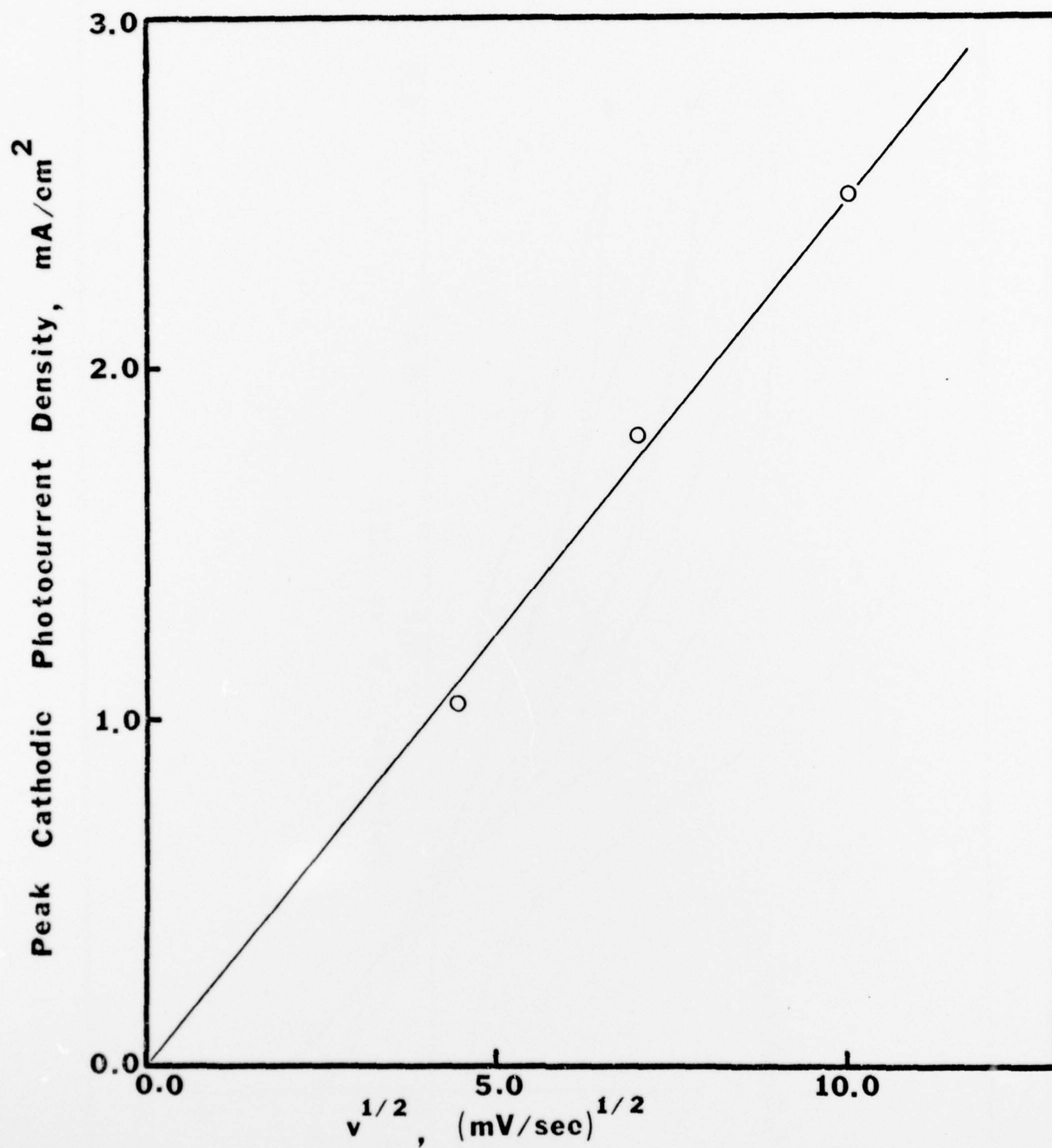
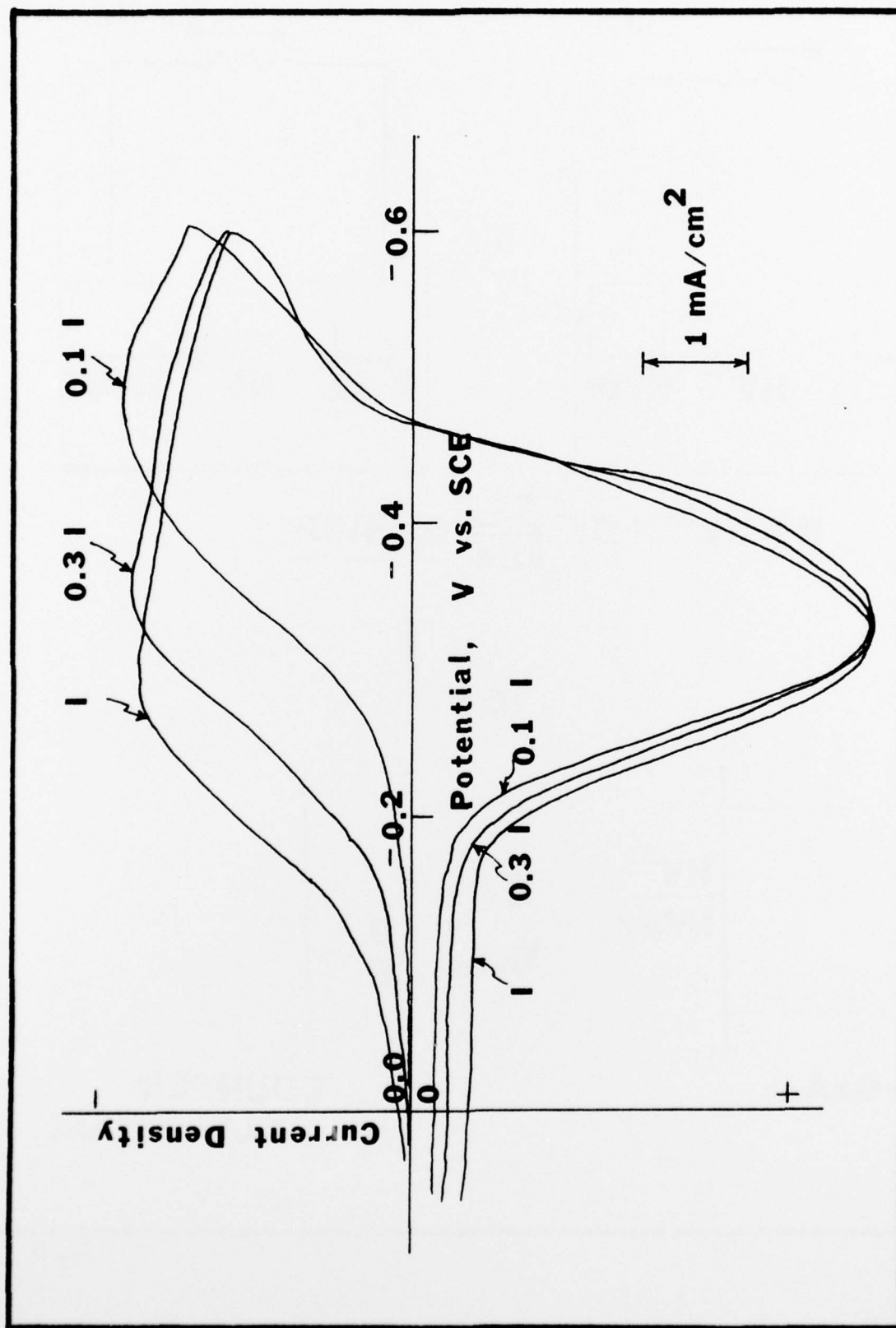


Fig 5



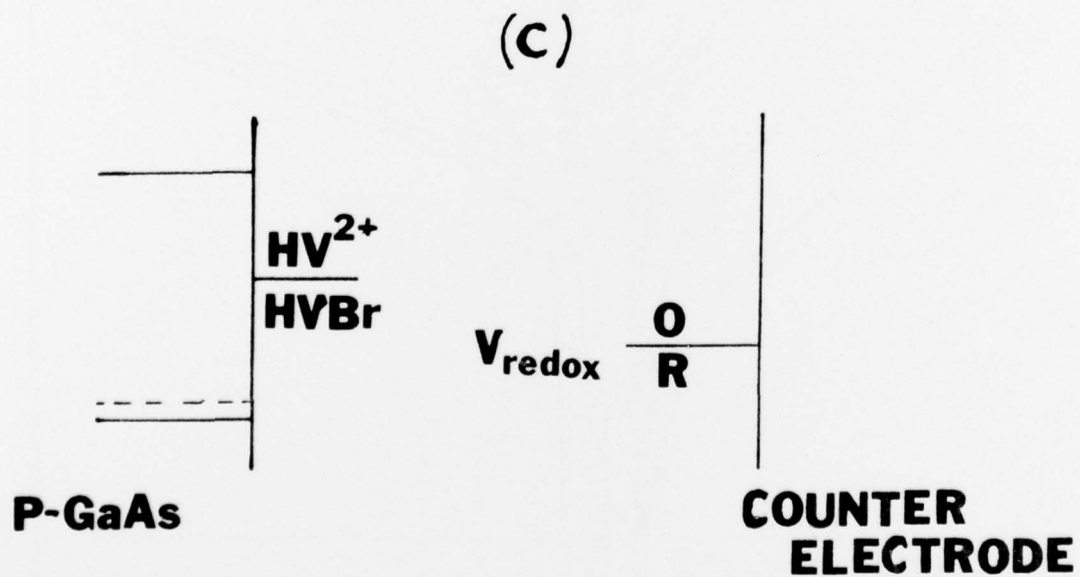
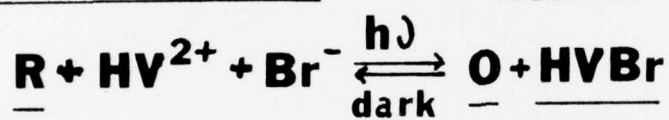
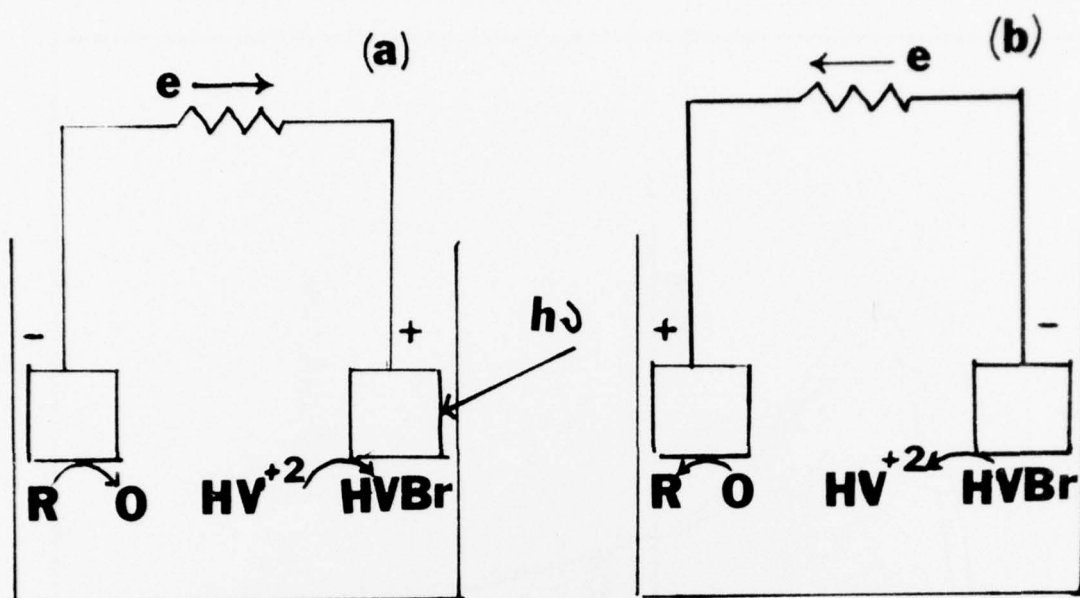


Fig 6



Fig 7

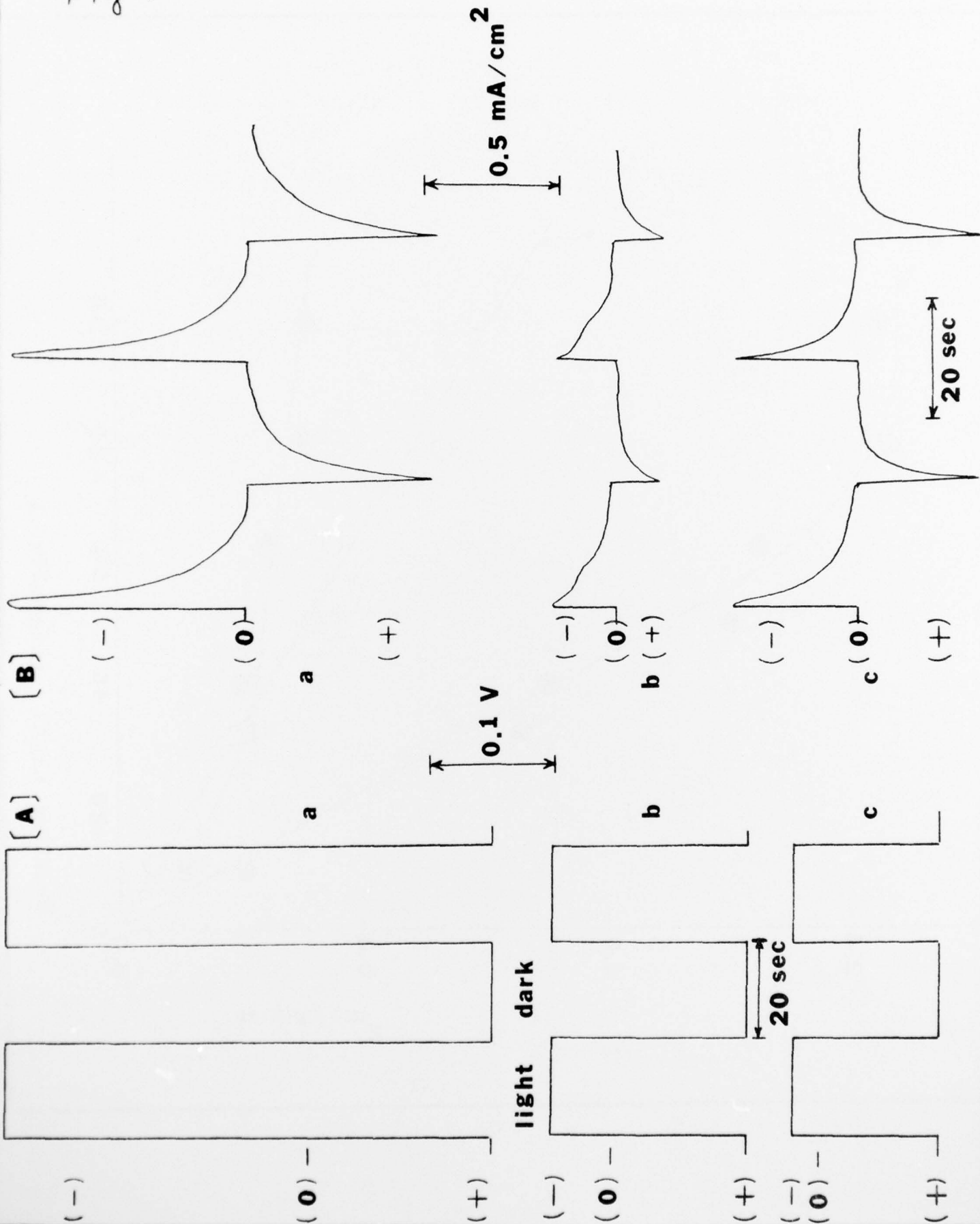


Fig 8

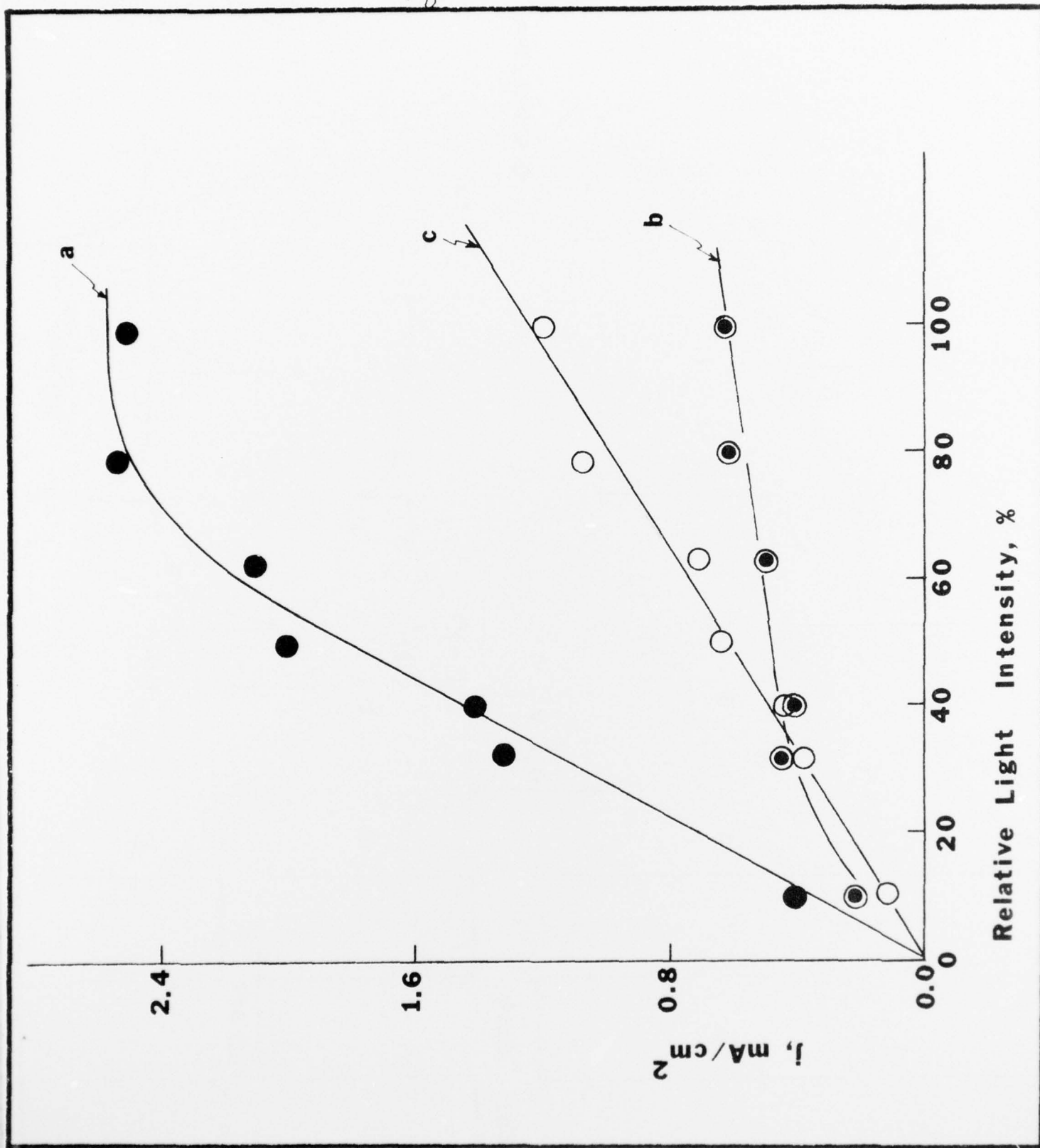
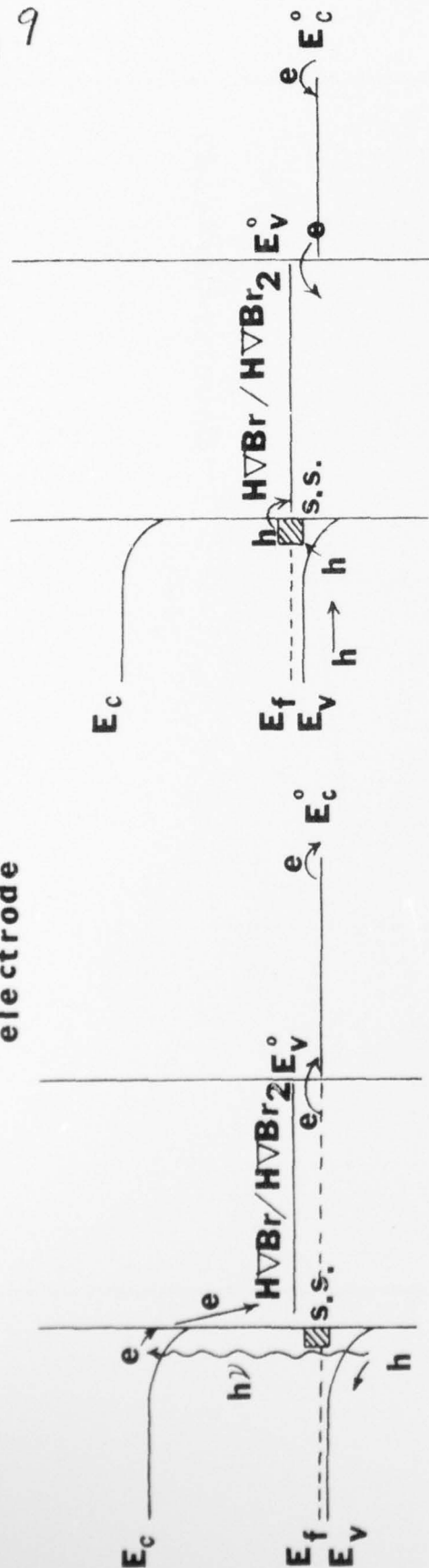
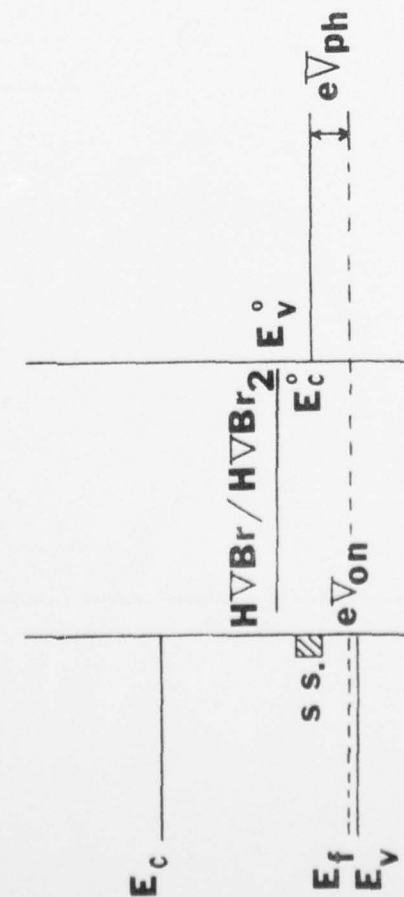


Fig 9

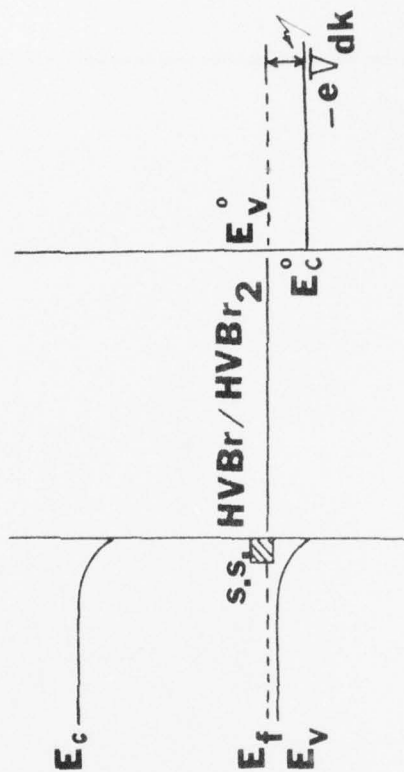
p-GaAs Electrolyte Counter-electrode



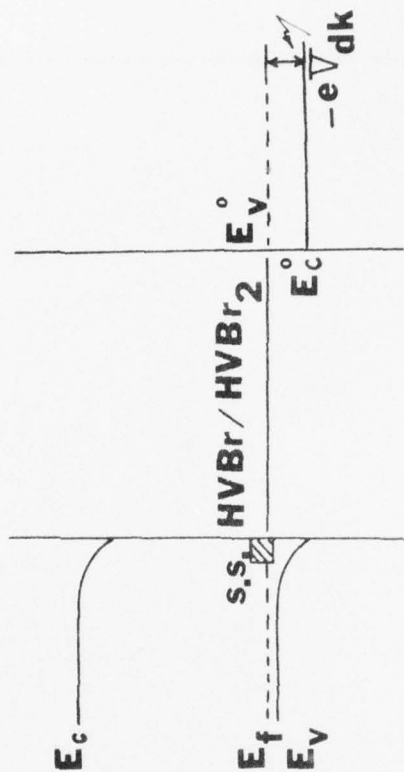
[a]



[b]

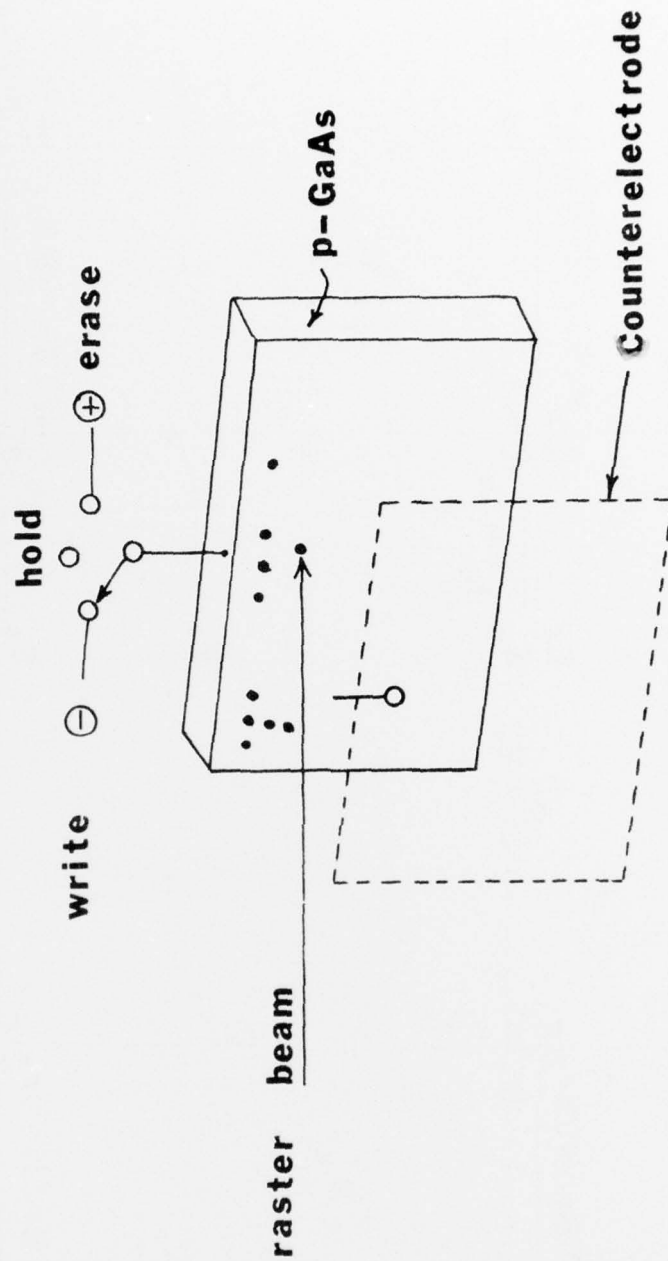


[c]



[d]

Fig 10



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